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DFT-Mbj Study of Electronic and Magnetic Properties of Cubic Cecro₃ Compound: An Ab-Initio Investigation

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Abstract

By considering density functional theory (DFT) in terms of ab-initio investigation, we have explored the structural, electronic and magnetic properties of cubic CeCrO₃ for the first time. In order to determine the structural stability of cubic CeCrO₃ compound, we optimized the structure of CeCrO₃ in non-magnetic (NM), ferromagnetic (FM) and Anti-ferromagnetic (AFM) phases by using PBE generalized gradient approximation (GGA) functional to find the exchangecorrelation potential. From structural optimization, the FM phase of CeCrO₃ is observed to be stable. For computing electronic and magnetic properties, the lately advanced modified Becke and Johnson local (spin) density approximation (mBJLDA) is used. Calculated band structures and density of states plots with an integer magnetic moment of 4 µB and reveal half-metallic character. In addition, s–d exchange constants (N₀α) and p–d exchange constant (N₀β) are determined, which are in agreement with a distinctive magneto-optical experiment.

Keywords: ab-initio calculations, cubic perovskite, half-metallic ferromagnetic, magnetic properties

Introduction

Magneto-electric (ME) characteristic in the multi-ferroics, now a day's has become of vigorous importance, due to the existence of correlated ordering parameter of electric and magnetic components that gets so much attention for famous existing application such as sensor of magnetic field, memory elements for multiple state, spintronic and multi-range microwave devices etc [1-3]. Detail information of crystal and magnetic structure is matter of great concern for useful application. Recently, the rare-earth ortho-ferrites have been reported like GdFeO₃ [5] and DyFeO₃ [4] have been reported for ferro-electricity as well as ME coupling effects. It is very difficult to perform polarization measurement caused by high leakage current due to high Neel temperature of RFeO₃ (T_N^{Fe} =620–740K). At the same time as perovskite chromite RCrO₃ exhibits magnetic properties at lower TN values (110–280K).RCrO₃ (R=Sm- Gd- Tb- Er-Tm- and Y) [6,7] was reported to show signs of a fairly great electric polarization



 $(0.2-0.8\mu C/cm^2)$, initially, at somewhat high temperatures equivalent to the T_N of the Cr sub-system. Additionally, LaCrO₃, CeCrO₃ have the maximum T_N in RCrO₃ compounds, which is advantageous for device applications at room temperature [8].

Perovskite oxides get so much attention due to its effective use in gas separation membranes, solid fuel cell and piezoelectric etc. [9-14]. Lanthanide doped perovskite type oxides, for example LnMeO₃ (Ln: lanthanides, Me: transition metals), have been accustomed for functional inorganic materials having a inclusive diversity of applications for alkaline fuel cells electrodes [15], gas ions sensors [16] and catalysts for fast and complete oxidation/reduction of CO, NO and other hydrocarbons [17]. Various fascinating physical properties such as structural, electronic, optical and magnetic properties are inter-dependent in transition-metal oxides [18, 19]. These materials are predictable for spintronics devices effectively.

The association of magnetic and electronic played a role to develop the research in area of spintronics. Half-metallic ferromagnetism has a significant part because of its spin- polarization at the Fermi-level which is necessary for the better performance of spintronic applications [20, 21]. All these points force us to explore half-metallic ferromagnets with important magnetic moment that wellmatched with existing semiconductor technology. Material similar to half-metal have a unique property to act as conductor in one direction of spin and insulator for opposite direction, therefore, it is very suitable for device applications and get attention for researcher that are working in the area of spintronic devices [22-24]. Some major examples are magnetic disk drives, magnetic tunnel junctions, magnetic hybrid technology for CMOS and magnetic sensor [25, 26], non-volatile magnetic random access memories (MRAM) [27, 28]. By getting research inspiration from de Groot et al. work, that explicate the insight of half-metallicity to compute band structure using the half-Heusler alloys NiMnSb [29]. Several research groups perform numerous experimental and computational studies on the HM ferromagnets, and many HM materials have been predicted and experimental verified [30].

CeCrO₃ belongs to Pm-3m (No. 221) space group and have cubic crystal structure. The atoms arrangement are as that Cr ions are placed at center of unit cell and coordinated with 6 oxygen ions, Ce ions are distributed at the corner of the cell and oxygen at center of the faces of unit cell. We have applied mBJ scheme, as introduced by Becker and Johnson (BJ) [31], as it can properly find electronic and magnetic characteristics. The structural properties of aforesaid crystals are calculated at ground state and equated with the prevalent theoretical and experimental data.



2. Method of Calculations

In this study, predicted results were obtained by carrying the Density Functional Theory (DFT), which is quantum mechanical approach that successful in predicting fundamental properties of compounds and alloys in terms of semiconducting trend. We employed DFT based full potential linearized augmented plane wave plus local orbital (FP-LAPW+lo) method within the frame work of Wien2K code [32]. In order to determine the ground state factors like lattice constant and bulk modulus, we used generalized gradient approximation (GGA) functional by considering the exchange-correlation potential suggested by Perdew, Burke, and Ernzerhof (PBE) [33]. Whereas, recently developed modified Becke-Johnson local density approximation functional (mBJLDA) [31] were used to analyze the magnetic and electronic properties. The aim of using mBJLDA potential for electronic properties is because of that exploring the improved predicting bandgap as associated with standard LDA [34] or GGA [33].

In FP-LAPW+lo method, ion cores inside non-overlapping spheres and a region of constant potential (interstitial region) are considered to the region between the spheres. In interstitial region, a plane wave expansion is used, whereas basis functions, potential and charge density were prolonged as arrangements of spherical harmonic functions. The value of lmax=10 in muffintin spheres for charge density and non-spherical potential was accomplished. For energy merging, basis function expand upto $R_{MT} \times K_{max} = 8$ (in the plane wave extension R_{MT} represent the minimum sphere radius and K_{MAX} the amount of the largest K vector). A mesh of 56 k-points was used in the irreducible part of the Brillouin zone (BZ) for structural, magnetic and electronic properties, which certifies the convergence, are 3000 k-points. In addition, the charge density was Fourier prolonged up to $G_{max} = 16$. For cubic CeCrO₃ perovskites, the R_{MT} values were elected to be 2.5, 1.72 and 1.87a.u. (atomic units) for Ce, Cr and O correspondingly. For energy convergence, the calculations of self-consistent were performed iteratively, when the total energy of the system is steady within 0.01 mRy.

3. Results and Discussion

3.1. Structural properties

To understand DFT based cubic CeCrO₃, structure stability is check by performing structural optimization in NM, FM and AFM phases. By using GGA-PBE scheme, optimization is done by minimizing the total energy with reverence to unit cell volume in each phase. From computed results (see Figure 1), the total energy difference between these two as $\Delta E = E_{NM} - E_{FM}$ and $\Delta E = E_{AFM} - E_{FM}$. The positive of ΔE confirmed that CeCrO₃ is stable in FM phase (see Table 1). In ABO₃ oxides with B=Mn, Cr, Fe, Ni and Co, most of such oxides have FM



character. Similar, FM character is also proved in our study. Therefore, in first step, we performed optimization in FM phase to compute the lattice constant a(Å)and bulk modulus B for CeCrO₃are shown in Table1.

The calculated tolerance factors for CeCrO₃ are mentioned in Table1. Our calculated value of tolerance factor is in adjacent covenant with the calculated results [35, 36]. In cubic perovskite, the tolerance factor lies between 0.93 and 1.02 [36] and our measured values employed in this range, illuminating the cubic structure of CeCrO₃ compounds. The bond lengths are measured between various atoms of the CeCrO₃ and also listed in Table1. The tolerance factor can be calculated using bond lengths by using the consequent formula:





Figure1. The computed total minimum energy versus unit cell volume in non-magnetic, ferro-magnetic and anti-ferromagnetic cubic CeCrO₃ compound

Table1.

Calculated lattice parameters a(Å), Bulk moduli B(GPa), Tolerance factor, Bond length, bandgap Eg (eV), Half-metallic $E_{HM}(eV)$, magnetic moments (μ_B) and exchange constant parameters of FM cubic CeCrO₃

Parameter	CeCrO ₃
$a_0(\text{\AA})$	3.877
B ₀ (GPa)	183.81
Tolerance factor	0.999
Bond length Cr-O	1.9373
Bond length Ce-O	2.7397
Bond length Cr-Ce	3.355
Eg (eV)	2.89
E _{HM} (eV)	0.38

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Total (µ _B)	4.0004
$Cr(\mu_B)$	2.5283
$Ce(\mu_B)$	0.9831
O (µ _B)	0.0642
$\Delta x (d)$	4.25
$\Delta x (pd)$	3.14
$\Delta E_{\rm C} (eV)$	0.42
$\Delta E_v (eV)$	2.70
$N_o \alpha$	0.33
N _o β	2.14

3.2. Electronic properties

Electronic band structure with mBJ DFT studies was discussed in this electronic part. Energy eigen values obtained with the help of KS equation for Electronic band structure. In the magnetic properties of CeCrO₃ show match of band structure due to spin up (\uparrow) and down (\downarrow) orientation of FM CeCrO₃ polarized band structure in Figure2. In the analysis of spin-up state band structure, it will be seen that the valance band (VB) maxima and conduction band (CB) minima both found at point M of Brillion Zone (see in Figure2). For above analysis, it is seen that up spin state of CeCrO₃ demonstrate the FM semiconductors behavior. The VB maxima cross the Fermi level (EF) for spin down channel in CeCrO₃, which describe the half-metallic manners is there for spin down (\downarrow) state. To disclose the starting point of density of states, partial density of state for CeCrO₃ is calculated and shown in the Figure3.



Figure 2. The Calculated Spin Polarized Ferromagnetic Band Structures for Cubic Cecro₃ Compound with Mbj Potential

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The (a) for majority spin (\uparrow) and the (b) for minority spin (\downarrow)

It understood able the Fermi level E_F is cross for spin up, total DOS, on the other hand the down channel form band gap at E_F , resultant the charges make a complete spin polarization and form this compound to utilize it for spintronics devices. For compound CeCrO₃ and by means of mBJ 2*p* states and 3*d* states of O and Cr contributed primarily in the region of VB involving -4 eV to Fermi level. If we deeply analyzed DOS from E_F to 1 eV, we observed the 4*f* orbitals of Ce contribution in DOS, after active over 3*d* states of Cr is clearly depicted. The shift in these states to-words high energies and the contribution of 4*f* states of Ce increments in the region of Fermi level clues to observed p-type conductivity and 100% spin polarization.



Figure 3. The calculated (a) total density of states (DOS), along partial DOS of (b) Ce, (c) Cr and (d) O by using mBJ for CeCrO₃ compound.

3.3. Magnetic Properties

We obtained magnetic moment CeCrO₃ from mBJ and listed in the Table1. Although, the calculated value is to some extent lesser than the theory value ($\mu_{Cr}=2.52 \ \mu$ B). The fact of the smaller value is that the electrons in 3*d* are



not totally confined but hybridized with oxygen of its 2p states. We have been also calculated magnetic moments of Ce ions. From the calculation we suggest that the total magnetic moment is predominantly because of Cr atoms and the M_{Ce} may be negligible. The half metallic characteristic of materials is verified by the numeral value of total magnetic moment. In our verdict the dual exchange method is liable for ferromagnetism seen in CeCrO₃ cubic perovskites and has indirect exchange interface among transition metals and rare-earths by the mean of anion O.

In addition splitting energy $\Delta_x(d)$ values of spin exchange that describes the role of 3*d* states of transition metal in exchange method is studied for Cr, and listed in the Table1. Most exciting calculated parameters from *p*-*d* and *s*-*d* coupling are swap coefficients N₀ α , N₀ β , correspondingly [37], that decide the swap interaction among TM *d*-state as well as charge carriers (holes and electrons in the valance band and conduction band respectively). The swap coefficients values for CeCrO₃ are shown in Table1. Additional, the value of N₀ α has lower than N₀ β indicate that the *s*-*d* contact at CB minima is greatly feebler than *p*-*d* interaction at VB maxima, that may be evidence of ferromagnetic behavior present in the given compound.

4. Conclusion

In conclusion, we have perceived structural, electronic and magnetic properties of FM CeCrO₃ compound by means of ab-initio calculations. To verify the stability of CeCrO₃ compound, we have optimized the structure in PM, FM and AFM phases and have computed their total energy differences (ΔE_1 and ΔE_2). The calculated value of ΔE_1 and ΔE_2 are positive, which confirm that CeCrO₃ is stable in FM phase. By analyzing band structure and density of state plots, we find that CeCrO₃ is a half-metallic ferromagnet, while predicted value of total magnetic moments is 4 µB. In addition, Cr 3*d* (unfilled) state due to *p*-*d* hybridization results in a fall of magnetic) ions. Furthermore, calculated exchange constant N₀ α and N₀ β indicate a lower value of N₀ α than N₀ β , which represents that spin-down state is more operative, due to *p*-*d* interaction at valance band maxima and ferromagnetism is confirmed by their strong hybridization. Furthermore, double-exchange mechanism is used to discuss the origin of the ferromagnetism in CeCrO₃.

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